## Letter

# <sup>155</sup>Gd Mössbauer effect and magnetic properties of GdMn<sub>6</sub>Ge<sub>6</sub>

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#### **Abstract**

The magnetic properties of GdMn<sub>6</sub>Ge<sub>6</sub> have been studied by magnetic measurements and <sup>155</sup>Gd Mössbauer spectroscopy. The Mn sublattice orders ferromagnetically in high magnetic fields and at high temperatures but in low fields and low temperatures there is a tendency to antiferromagnetic ordering. Antiferromagnetic order was also found in YMn<sub>6</sub>Ge<sub>6</sub>. The electric field gradient derived from the quadrupolar splitting of the <sup>155</sup>Gd Mössbauer spectra is substantially larger than in the isotypic compound GdMn<sub>6</sub>Sn<sub>6</sub>.

Rare earth compounds of the type RMn<sub>6</sub>Sn<sub>6</sub> have been reported recently to possess quite interesting magnetic properties owing to the presence of a magnetic moment on the Mn atoms [1, 2]. The magnetic ordering temperatures are well above room temperature but the magneto-crystalline anisotropy is only of moderate magnitude. <sup>155</sup>Gd Mössbauer spectroscopy, used as a probe of the local electric field gradient, has revealed that this may be due to the relatively small size of the second order crystal field parameter A<sub>2</sub><sup>0</sup> associated with the crystal field induced rare earth sublattice anisotropy [2].

In the present study we have focussed our attention on the properties of the compound GdMn<sub>6</sub>Ge<sub>6</sub>. Several compounds of the series RMn<sub>6</sub>Ge<sub>6</sub> were prepared by us and found to crystallize in the same structure type as adopted by RMn<sub>6</sub>Sn<sub>6</sub>. This structure type (HfFe<sub>6</sub>Ge<sub>6</sub>) is relatively simple and comprises only a single R site and a single 3d site, the coordination of the R sites being similar but not identical to that found in the CeCo<sub>3</sub>B<sub>2</sub>structure type [3].

The GdMn<sub>6</sub>Ge<sub>6</sub> sample was prepared by arc melting starting materials with purities of at least 99.9%. After

arc melting, the sample was wrapped in Ta foil, sealed in an evacuated quartz tube and then vacuum annealed at  $800^{\circ}$ C for about 4 weeks. X-ray diffraction showed the sample to be approximately single phase after annealing. The lattice constants are a=0.52371 nm and c=0.81828 nm. In order to determine the easy magnetization direction in GdMn<sub>6</sub>Ge<sub>6</sub> X-ray diagrams were also taken from magnetically aligned powder. Conclusive evidence was obtained that the easy magnetization direction in GdMn<sub>6</sub>Ge<sub>6</sub> is perpendicular to the c axis.

Results of magnetic measurements, made on a SQUID magnetometer in a field of 0.5 T, are shown in Fig. 1. Magnetic ordering is seen to occur at  $T_{\rm c}=490$  K. In the range between 200 and 400 K the magnetization reaches a value of about 30 Am<sup>2</sup> g k<sup>-1</sup> (30 emu g<sup>-1</sup>) which corresponds to about 5  $\mu_{\rm B}$  per formula unit GdMn<sub>6</sub>Ge<sub>6</sub>. Below about 200 K a dramatic decrease in the magnetization occurs, leading to a magnetic moment of only about 1  $\mu_{\rm B}$  per formula unit at 4.2 K. These results may be compared with those obtained on YMn<sub>6</sub>Ge<sub>6</sub>. As seen in Fig. 1 this compound gives rise to antiferromagnetic ordering at  $T_{\rm N}=485$  K.

The magnetic isotherm of  $GdMn_6Ge_6$  at 4.2 K was measured at the Amsterdam high-field installation [4] on fine powder particles that were allowed to orient themselves freely in the applied field. The results are shown in Fig. 2. In the low-field region the magnetization rises very rapidly. Above about 10 T the magnetization varies linearly with the field strength. Extrapolation to H=0 leads to a saturation moment of 5.0  $\mu_B$  per formula unit. Under the assumption of antiparallel

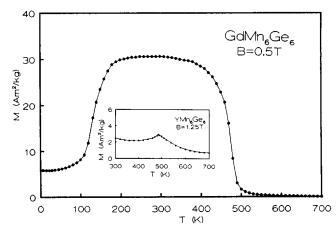


Fig. 1. Temperature dependence of the magnetization in GdMn<sub>6</sub>Ge<sub>6</sub> and YMn<sub>6</sub>Ge<sub>6</sub> (inset) measured in fields of 0.5 T and 1.25 T respectively.

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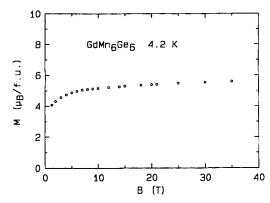


Fig. 2. Field dependence of the magnetization of GdMn<sub>6</sub>Ge<sub>6</sub> at 4.2 K.

coupled Gd and Mn sublattice magnetizations the saturation moment of 5.0  $\mu_{\rm B}$  may be interpreted as being the result of an Mn sublattice moment of 12.0  $\mu_{\rm B}$  and a Gd sublattice moment of 7.0  $\mu_{\rm B}$  (free ion value). The shape of the magnetic isotherm suggests that the situation is different in low fields since one would have expected that the magnetization has a linear field dependence in fields much lower than 10 T for ferromagnetic particles able to rotate freely in the sample holder. Most likely the Mn sublattice is no longer ferromagnetic in zero field but rather antiferromagnetic, leaving only a relatively small net Mn sublattice moment induced by the molecular field of the Gd sublattice. As may be seen from Fig. 1 this situation is present only up to about 100 K. The strong rise in the magnetization between 100 and 200 K suggests that with increasing temperature there is a transition from an antiferromagnetic Mn sublattice to a ferromagnetic Mn sublattice. This transition is most likely driven by thermal expansion. This hypothesis is based on the fact that Mn atoms have a tendency to couple antiparallel for small Mn-Mn distances but to couple parallel for sufficiently large Mn-Mn distances. It agrees with preliminary observations on TbMn<sub>6</sub>Ge<sub>6</sub> where the corresponding transition occurs at substantially higher temperatures, only slightly below T<sub>c</sub>. More experiments are currently being undertaken to study the origin of this interesting behaviour in more detail.

The Mössbauer spectrum of GdMn<sub>6</sub>Ge<sub>6</sub> was obtained by means of the 86.5 keV resonance of <sup>155</sup>Gd. The source consisted of neutron-irradiated SmPd<sub>3</sub> prepared with samarium enriched to 98% in <sup>154</sup>Sm. Details of the spectrometer are given elsewhere [5]. The spectrum obtained is shown in Fig. 3.

We analysed the spectrum by means of a least-squares fitting procedure based on the diagonalization of the full nuclear Hamiltonian and used a transmission integral. The independently refined variables consisted of the isomer shift (IS), the effective hyperfine field  $(H_{\text{eff}})$  and the quadrupole splitting (QS) (or the electric

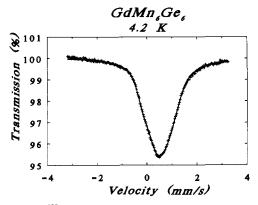


Fig. 3. <sup>155</sup>Gd Mössbauer spectrum of GdMn<sub>6</sub>Ge<sub>6</sub> at 4.2 K. The solid curve through the data points represents a fit.

TABLE 1. Hyperfine parameters derived from fitting the <sup>155</sup>Gd Mössbauer spectra at 4.2 K of GdMn<sub>6</sub>Ge<sub>6</sub> and GdMn<sub>6</sub>Sn<sub>6</sub>

Compound	$V_{zz} (10^{21} \text{ V m}^{-2})$	B <sub>eff</sub>   (T)	IS (mm s <sup>-1</sup> )	θ (deg)	η
GdMn <sub>6</sub> Ge <sub>6</sub> GdMn <sub>6</sub> Sn <sub>6</sub>			$0.49 \pm 0.01$ $0.55 \pm 0.001$	90 90	0

field gradient tensor element  $V_{zz'}$  obtained via the relation QS =  $(1/4)eQV_{zz}$   $(3\cos^2\theta - 1)$ , where the value  $Q = 1.30 \times 10^{-28} \text{ m}^2$  was taken from Tanaka et al. [6]. The fitting procedure has furthermore been performed with the constraint that the angle  $\theta$  between  $H_{\rm eff}$  and the c-axis be 90° as was derived from results obtained previously for the isotypic compound Gd-Mn<sub>6</sub>Sn<sub>6</sub>. The absorber and source line widths were constrained to 0.25 and 0.36 mm s<sup>-1</sup> for the transmission integral. The hyperfine parameters corresponding to the best fit are listed in Table 1, where they can be compared with the hyperfine parameters obtained previously [2] for the isotypic compound GdMn<sub>6</sub>Sn<sub>6</sub>. It can be seen from Table 1 that the electric field gradient  $V_{zz}$  at the nuclear Gd site has increased by more than 60% compared with GdMn<sub>6</sub>Sn<sub>6</sub>.

The electric field gradient in GdMn<sub>6</sub>Sn<sub>6</sub> was discussed extensively in a previous report [2] where a comparison was made between the crystal structures and the expected  $V_{zz}$  values in CeCo<sub>3</sub>B<sub>2</sub> type structures and HfFe<sub>6</sub>Ge<sub>6</sub> type structures. Both structure types have in common that the central minority atom is surrounded in the equatorial plane by a hexagon of six s,p atoms while in the plane below and above the equatorial plane there are hexagons consisting of transition metal atoms. For Gd-based compounds this coordination is able to produce large asphericities of the 6p and 5d on-site valence electrons of Gd which in turn produces a large value of  $V_{zz}$  [7, 8]. However, in the HfFe<sub>6</sub>Ge<sub>6</sub>type structure there are two additional near neighbour s,p atoms along the c direction which have a strongly detrimental influence on the on-site valence electron Letter L31

asphericities of Gd, explaining the comparatively low value of  $V_{zz}$  in GdMn<sub>6</sub>Sn<sub>6</sub>. This detrimental influence on  $V_{zz}$  is expected to be larger the lower the electron density at the atomic cell boundaries of the s, p element [9]. Given the fact that the latter quantity is lower for Sn than for Ge [10] one may conclude that the detrimental influence of the two s, p-neighbour atoms in the Gd coordination shell is less severe in GdMn<sub>6</sub>Ge<sub>6</sub> than in GdMn<sub>6</sub>Sn<sub>6</sub>, explaining the higher value of  $V_{zz}$  in the former compound compared with the latter. Further investigations are planned to study the correlations between the  $V_{zz}$  values found in GdMn<sub>6</sub>Ge<sub>6</sub> and GdMn<sub>6</sub>Sn<sub>6</sub> and the corresponding rare earth sublattice anisotropies in the series RMn<sub>6</sub>Ge<sub>6</sub> and RMn<sub>6</sub>Sn<sub>6</sub>.

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